# Synthesis and characterization of new vanadates of antimony, $\mathrm{ASbV}_{2} \mathrm{O}_{8}(\mathrm{~A}=\mathrm{K}, \mathrm{Rb}, \mathrm{Tl}$ or Cs$) \dagger$ 

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Four new vanadates of antimony, $\mathrm{KSbV}_{2} \mathrm{O}_{8} \mathbf{1}, \mathrm{RbSbV}_{2} \mathrm{O}_{8} \mathbf{2}, \mathrm{TlSbV}_{2} \mathrm{O}_{8} \mathbf{3}$ and $\mathrm{CsSbV}_{2} \mathrm{O}_{8} \mathbf{4}$, have been prepared by solid state synthesis and structurally characterized by single crystal X-ray diffraction. Compounds $\mathbf{1}$ to $\mathbf{3}$ have a three-dimensional $\left(\mathrm{SbV}_{2} \mathrm{O}_{8}{ }^{-}\right)_{\infty}$ anionic framework with one dimensional channels occupied by counter cations. Compound $\mathbf{4}$ has two dimensional $\left(\mathrm{SbV}_{2} \mathrm{O}_{8}^{-}\right)_{\infty}$ layers interleaved with $\mathrm{Cs}^{+}$ions. Both types of anionic frameworks are built from corner connected $\mathrm{SbO}_{6}$ octahedra and $\mathrm{VO}_{4}$ tetrahedra. The spectroscopic data are briefly discussed.

## Introduction

Solid compounds possessing structural frameworks with well defined tunnels and layers are extensively investigated because their unique and complex structural chemistry endows them with interesting chemical and physical properties. Phosphates in the A-M-P-O quaternary system form one such family of compounds showing rich structural chemistry with anionic frameworks built from $\mathrm{MO}_{6}$ octahedra and $\mathrm{PO}_{4}$ tetrahedra and $\mathrm{A}^{n+}$ ions as counter cations. ${ }^{1-13}$ Research on the synthetic and structural chemistry of these materials continues to be pursued with the idea of obtaining them with accessible redox catalytic activity, anisotropic electrical conductivity and ion exchange properties. For instance, a series of phosphates with general formula $\mathrm{AMP}_{2} \mathrm{O}_{8}(\mathrm{~A}=\mathrm{K}, \mathrm{Rb}, \mathrm{Cs} \text { or } \mathrm{Tl} ; \mathrm{M}=\mathrm{Nb}, \mathrm{Ta} \text { or } \mathrm{Sb})^{8-13}$ exhibit interesting properties like ion exchange and luminescence. These compounds have been shown to possess three types of structures, namely, layered $\alpha-\mathrm{ZrP}$ type, ${ }^{14}$ with $\left[\mathrm{M}\left(\mathrm{PO}_{4}\right)_{2}\right]^{-}$anionic framework, layered and three dimensional structures with $\left[\mathrm{MO}\left(\mathrm{P}_{2} \mathrm{O}_{7}\right)\right]^{-}$anionic framework containing pyrophosphates. One would expect similar structural diversity in isomorphous vanadates containing $\mathrm{VO}_{4}$ tetrahedra, but surprisingly we have found that not many binary vanadates with extended structures have been reported. Such vanadates, when compared to phosphates, could be expected to show enhanced redox activity. We have, therefore, initiated a project on the synthetic and structural chemistry of new vanadates isomorphous with the phosphates. Here we report the synthesis and characterization of four new vanadates, $\mathrm{ASbV}_{2} \mathrm{O}_{8}(\mathrm{~A}=\mathrm{K}$, $\mathrm{Rb}, \mathrm{Tl}$ or Cs ), with two types of structures.

## Results and discussion

The four $\mathrm{ASbV}_{2} \mathrm{O}_{8}$ compounds, $\mathrm{M}=\mathrm{K} \mathbf{1}, \mathrm{Rb} \mathbf{2}, \mathrm{Tl} 3$ or $\mathrm{Cs} \mathbf{4}$, could be prepared by solid state synthesis at temperatures as low as $600^{\circ} \mathrm{C}$. Our attempts to grow single crystals of $\mathbf{1}$, which has a low melting point of $700^{\circ} \mathrm{C}$, by slow cooling of melt were not successful. Only the chemical vapour transport technique, using the transporting agent $\mathrm{SbCl}_{5}$, enabled the isolation of these compounds in the form of single crystals suitable for X-ray diffraction. The pertinent crystallographic data and the data collection parameters of these four compounds are summarized in Table 1. The powder X-ray diffraction (XRD) patterns (SUP 57412) of these compounds compare well

[^0]with those simulated based on the single crystal structure using the LAZY PULVERIX ${ }^{15}$ program and thus establish the monophasic nature of the polycrystalline samples. Our attempts to prepare $\mathrm{NaSbV}_{2} \mathrm{O}_{8}$ were not successful.

## Crystal structure

Compounds 1, 2 and 3 have a three-dimensional anionic framework $\left(\mathrm{SbV}_{2} \mathrm{O}_{8}{ }^{-}\right)_{\infty}$ containing one-dimensional channels occupied by the counter cations $\mathrm{K}^{+}, \mathrm{Rb}^{+}$and $\mathrm{Tl}^{+}$. While $\mathbf{1}$ and $\mathbf{2}$ are isostructural, crystallizing in the space group Pnma, $\mathbf{3}$ crystallizes in the space group $P 2_{1} / \mathrm{c}$ with $\beta$ angle very close to $90^{\circ}$ (Table 1).

The three-dimensional $\left(\mathrm{SbV}_{2} \mathrm{O}_{8}{ }^{-}\right)_{\infty}$ anionic framework of these three compounds is built from corner sharing of $\mathrm{SbO}_{6}$ octahedra and $\mathrm{VO}_{4}$ tetrahedra. Each $\mathrm{SbO}_{6}$ octahedron is corner connected to six $\mathrm{VO}_{4}$ tetrahedra. Each $\mathrm{VO}_{4}$ tetrahedron, on the other hand, is corner connected to only three $\mathrm{SbO}_{6}$ octahedra. The three-dimensional nature of the framework can be conceived as follows. First, each $\mathrm{SbO}_{6}$ octahedron, with its two trans corners unshared, is corner connected to four $\mathrm{VO}_{4}$ tetrahedra and each $\mathrm{VO}_{4}$ tetrahedron connects two such octahedra to form a one-dimensional $\mathrm{SbV}_{2} \mathrm{O}_{10}$ chain as shown in Fig. 1. It is noteworthy that the edge connecting the unshared oxygen atoms of each tetrahedron in one row is approximately perpendicular to that of the tetrahedron in the other row. In other words, one row of the tetrahedra in the chain is oriented perpendicular to the other row. Two such chains are laterally fused, whereby one row of tetrahedra of each chain is corner connected to octahedra of the other chain to form a onedimensional ribbon of composition $\mathrm{Sb}_{2} \mathrm{~V}_{4} \mathrm{O}_{18}$ as shown in Fig. 1. Such ribbons are connected to one another by further sharing of corners of octahedra with tetrahedra to form a three-dimensional framework with one-dimensional channels as shown in the unit cell diagram (Fig. 2) of $\mathrm{KSbV}_{2} \mathrm{O}_{8}$. It is evident from the figure that each tetrahedron points its unshared oxygen towards one-dimensional channels and these channels, parallel to the short ( $\approx 5.5 \AA$ ) $b$ axis, are occupied by monovalent counter cations. In the case of compound 3 the channels are parallel to the short $a$ axis. This type of three dimensional stucture has been reported for molybdates of indium, ${ }^{16,17} \mathrm{KInMo}_{2} \mathrm{O}_{8}, \beta-\mathrm{RbInMo} \mathrm{O}_{2} \mathrm{O}_{8}$ and $\beta$ - $\mathrm{CsInMo}_{2} \mathrm{O}_{8}$.

For compounds $\mathbf{1}$ and $\mathbf{2}$ there are one antimony atom, two crystallographically distinct vanadium atoms, $\mathrm{V}(1)$ and $\mathrm{V}(2)$, and six oxygen atoms in the asymmetric unit. The asymmetric unit of $\mathbf{3}$ and $\mathbf{4}$, however, contains two more oxygen atoms. The oxygen co-ordinations of antimony and vanadium atoms for $\mathbf{1}$, $\mathbf{3}$ and $\mathbf{4}$ are represented in Fig. 3 with the atom labelling scheme. The features of these polyhedra in $\mathbf{1 , 2}$ and $\mathbf{3}$ are the same and

Table 1 Crystal data and data collection parameters for compounds 1-4

| Formula | $\mathrm{KO}_{8} \mathrm{SbV}_{2} 1$ | $\mathrm{O}_{8} \mathrm{RbSbV}_{2} 2$ | $\mathrm{O}_{8} \mathrm{SbTlV}_{2} 3$ | $\mathrm{CsO}_{8} \mathrm{SbV}_{2} 4$ |
| :---: | :---: | :---: | :---: | :---: |
| M | 390.68 | 437.05 | 555.96 | 484.49 |
| Crystal system | Orthorhombic | Orthorhombic | Monoclinic | Monoclinic |
| Space group | Pnma | Pnma | $P 2{ }_{1} / \mathrm{c}$ | $P 2_{1} / n$ |
| alÅ | 14.6248(10) | 14.722(2) | 5.569(5) | 5.627(2) |
| b/Å | 5.5633(10) | 5.5679(8) | 8.670(4) | 16.178(4) |
| clÅ | 8.5168(10) | 8.6697(13) | 14.630(3) | 8.400(5) |
| $\beta 1{ }^{\circ}$ |  |  | 90.14(4) | 90.30(5) |
| $U 1 \AA^{3}$ | 692.9(2) | 710.7(2) | 706.4(7) | 764.6(6) |
| Crystal dimensions/mm | $0.09 \times 0.15 \times 0.3$ | $0.08 \times 0.12 \times 0.2$ | $0.07 \times 0.1 \times 0.25$ | $0.08 \times 0.12 \times 0.3$ |
| $D_{\mathrm{c}} / \mathrm{Mg} \mathrm{m}^{-3}$ | 3.745 | 4.085 | 5.228 | 4.209 |
| $F(000)$ | 720 | 792 | 968 | 864 |
| $2 \theta$ Range | 5.54-49.94 ( $+h,+k,+l)$ | 5.46-49.92 ( $+h,+k,+l)$ | 5.46-49.94 ( $\pm h,+k,+l)$ | 5.04-49.96 ( $\pm h,+k,+l)$ |
| $\mu(\mathrm{Mo}-\mathrm{K} \alpha) / \mathrm{mm}^{-1}$ | 7.078 | 12.918 | 28.215 | 10.563 |
| Maximum, minimum transmission factors | 0.92, 0.75 | 0.82, 0.50 | 0.82, 0.48 | 0.85, 0.48 |
| Total no. reflections | 676 | 700 | 1492 | 1550 |
| No. independent reflections ( $R_{\text {int }}$ ) | 676 (0.0000) | 700 (0.0000) | 1250 (0.0205) | 1345 (0.0551) |
| No. parameters refined | 67 | 68 | 105 | 109 |
| $R$ | 0.0254 | 0.0310 | 0.0261 | 0.0491 |
| $R^{\prime}$ | 0.0744 | 0.0779 | 0.0580 | 0.1262 |
| $x$ | 0.0248 | 0.0500 | 0.0375 | 0.0494 |
| $y$ | 7.11 | 2.65 | 0.24 | 36.51 |
| Goodness of fit | 1.411 | 1.163 | 1.155 | 1.186 |
| Largest shift/e.s.d. | 0.00 | 0.00 | 0.049 | 0.00 |
| Largest peak/e $\AA^{-3}$ | 1.042 | 1.037 | 1.477 | 2.742 |

Details in common: graphite-monochromated Mo-K $\alpha$ radiation ( $\lambda 0.71073 \AA$ ); 20- $\omega$ scans; $Z=4 ; R=\Sigma| | F_{\mathrm{o}}\left|-\left|F_{\mathrm{c}}\right|\right| \Sigma\left|F_{\mathrm{o}}\right| ; R^{\prime}=\left[\Sigma w\left(\left|F_{\mathrm{o}}\right|-\mid F_{\mathrm{c}}\right)^{2} /\right.$ $\left.\Sigma w \mid F_{\mathrm{o}}^{2}\right]^{\frac{1}{2}} ; w=1 /\left[\sigma^{2}\left(\left|F_{\mathrm{o}}\right|\right)^{2}+(x P)^{2}+y P\right] ; P=\left[\operatorname{Max}\left(F_{\mathrm{o}}^{2}, 0\right)+2 F_{\mathrm{c}}^{2}\right] / 3$.


Fig. 1 End-on view (top) and side view (bottom) of the $\mathrm{SbV}_{2} \mathrm{O}_{10}$ chain (left) and $\mathrm{Sb}_{2} \mathrm{~V}_{4} \mathrm{O}_{18}$ ribbon (right).
we discuss these features by taking $\mathbf{1}$ as an example. The $\mathrm{SbO}_{6}$ octahedron, corner connected to three $\mathrm{V}(1) \mathrm{O}_{4}$ and three $\mathrm{V}(2) \mathrm{O}_{4}$ tetrahedra, is nearly regular with antimony occupying the centre of the octahedron as evident from the values of the $\mathrm{Sb}-\mathrm{O}$ bond lengths, $\mathrm{O} \cdots \mathrm{O}$ non-bonding edges and bond angles given in Tables 2 and 3. The $\mathrm{V}(1) \mathrm{O}_{4}$ tetrahedra are involved in the fusion of $\mathrm{SbV}_{2} \mathrm{O}_{10}$ chains to form $\mathrm{Sb}_{2} \mathrm{~V}_{4} \mathrm{O}_{18}$ ribbons whereas $\mathrm{V}(2) \mathrm{O}_{4}$ tetrahedra connect such ribbons to form the three dimensional network. These two types of tetrahedra are oriented perpendicular to each other. The vanadium atom forms a shorter bond with an unshared oxygen atom, namely $\mathrm{O}(3)$ in $\mathrm{V}(1) \mathrm{O}_{4}$ and $\mathrm{O}(6)$ in $\mathrm{V}(2) \mathrm{O}_{4}$ tetrahedra, than with the other three oxygen atoms (Fig. 3). For 3 the unshared oxygen atoms are $\mathrm{O}(4)$ and $\mathrm{O}(8)$. The three longer V-O bonds have more or less the same length in $\mathrm{V}(1) \mathrm{O}_{4}$ tetrahedra but not in $\mathrm{V}(2) \mathrm{O}_{4}$
(Tables 2 and 3). The vanadium atoms occupy the centres of the tetrahedra and the difference between the minimum and maximum values of $\mathrm{O} \cdots \mathrm{O}$ distances is more for $\mathrm{V}(2) \mathrm{O}_{4}$ tetrahedra $(0.219 \AA)$ than for $\mathrm{V}(1) \mathrm{O}_{4}$ tetrahedra $(0.177 \AA)$, indicating a greater distortion in the former. This difference in distortion is reflected in the values of their bond angles as well. Among the isostructural compounds $\mathbf{1}$ and $\mathbf{2}$, the bigger $\mathrm{Rb}^{+}$ion seems to lead to a decrease in the average $\mathrm{Sb}-\mathrm{O}$ bond length with concomitant increase in the average $\mathrm{V}(1)-\mathrm{O}$ bond length.

The compound $4, \mathrm{CsSbV}_{2} \mathrm{O}_{8}$, has a layer structure consisting of two dimensional $\left(\mathrm{SbV}_{2} \mathrm{O}_{8}^{-}\right)_{\infty}$ layers interleaved with counter cations $\mathrm{Cs}^{+}$. The two-dimensional $\left(\mathrm{SbV}_{2} \mathrm{O}_{8}{ }^{-}\right)_{\infty}$ anionic framework is built, as in the three dimensional framework of $\mathbf{1}$ to $\mathbf{3}$, from corner sharing of $\mathrm{Sb}_{2} \mathrm{~V}_{4} \mathrm{O}_{18}$ ribbons but the mode of connection is different. The connectivity between these ribbons
extends in only one direction, so as to give a two dimensional anionic layer. As in the other three compounds, each $\mathrm{SbO}_{6}$ octahedron is corner connected to six $\mathrm{VO}_{4}$ tetrahedra and each $\mathrm{VO}_{4}$ tetrahedron is connected to three $\mathrm{SbO}_{6}$ octahedra. The


Fig. 2 Structure of $\mathrm{KSbV}_{2} \mathrm{O}_{8}$ with polyhedral representation of the $\left(\mathrm{SbV}_{2} \mathrm{O}_{8}^{-}\right)_{\infty}$ framework and potassium as filled circles. All atoms are represented by circles of arbitrary radii.


Fig. 3 The ORTEP ${ }^{18}$ plots of the $\mathrm{SbV}_{2} \mathrm{O}_{12}$ units in compounds $\mathbf{1}$ (top), 3 (middle) and 4 (bottom) showing the atom labelling scheme. Antimony in $\mathbf{3}$ is represented by a circle of arbitrary radius and all the other atoms with $50 \%$ probability.

Table 2 Bond lengths $(\AA)$ and selected angles $\left({ }^{\circ}\right)$ and $\mathrm{O} \cdots \mathrm{O}$ non-bonding distances in $\mathrm{ASbV}_{2} \mathrm{O}_{8}(\mathrm{~A}=\mathrm{K}$ or Rb$)$
$\mathrm{SbO}_{6}$ octahedra


Table 3 Bond lengths $(\AA)$ and selected angles $\left({ }^{\circ}\right)$ and $\mathrm{O} \cdots \mathrm{O}$ non-bonding distances in $\mathrm{ASbV}_{2} \mathrm{O}_{8}(\mathrm{~A}=\mathrm{Tl}$ or Cs$)$
$\mathrm{SbO}_{6}$ octahedra
$\mathrm{A}=\mathrm{Tl}$

| Sb | $\mathrm{O}(3)$ | $\mathrm{O}(2)$ | $\mathrm{O}(5)$ | $\mathrm{O}(1)$ | $\mathrm{O}(6)$ | $\mathrm{O}(7)$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: |
| $\mathrm{O}(3)$ | 1.963(4) | 2.682(5) | $2.855(6)$ | 2.792(6) | - | 2.841(5) |
| $\mathrm{O}(2)$ | 86.1(2) | 1.966(4) | $2.858(6)$ | 2.795(6) | $2.829(5)$ | - |
| $\mathrm{O}(5)$ | 93.1(2) | 93.2(2) | 1.969(4) | - | 2.746 (6) | 2.746(5) |
| $\mathrm{O}(1)$ | 90.5(2) | 90.5(2) | 175.0(2) | 1.969(4) | $2.719(5)$ | $2.722(5)$ |
| $\mathrm{O}(6)$ | 177.6(2) | 92.3(2) | 88.8(2) | 87.7(2) | 1.956(4) | 2.739 (6) |
| $\mathrm{O}(7)$ | 92.8(2) | 177.9(2) | 88.7(2) | 87.7(2) | 88.7(2) | 1.961(4) |
| $\mathrm{A}=\mathrm{Cs}$ |  |  |  |  |  |  |
| Sb | $\mathrm{O}(7)$ | $\mathrm{O}(6)$ | $\mathrm{O}(1)$ | $\mathrm{O}(3)$ | $\mathrm{O}(5)$ | $\mathrm{O}(2)$ |
| $\mathrm{O}(7)$ | 1.946(10) | 2.801(13) | 2.729(14) | - | 2.781(13) | $2.889(15)$ |
| $\mathrm{O}(6)$ | 90.5(4) | 1.997(9) | 2.847(13) | 2.674(14) | 2.752(13) | - |
| $\mathrm{O}(1)$ | 88.6(4) | 92.0(4) | 1.962(8) | 2.752(12) | - | 2.785(12) |
| $\mathrm{O}(3)$ | 175.9(4) | 85.8(4) | 89.9(4) | 1.932(9) | 2.752(12) | $2.712(12)$ |
| $\mathrm{O}(5)$ | 91.1(4) | 88.5(4) | 179.5(3) | 90.4(4) | 1.948(8) | 2.740 (12) |
| $\mathrm{O}(2)$ | 95.4(4) | 173.6(4) | 90.5(4) | 88.4(4) | 89.1(4) | 1.959(9) |

$\mathrm{VO}_{4}$ tetrahedra



Fig. 4 The $\left(\mathrm{SbV}_{2} \mathrm{O}_{8}{ }^{-}\right)_{\infty}$ layer in $\mathrm{CsSbV}_{2} \mathrm{O}_{8}$ viewed along the $b$ axis.
polyhedral representation of the $\left(\mathrm{SbV}_{2} \mathrm{O}_{8}^{-}\right)_{\infty}$ layer viewed along the $b$ axis is given in Fig. 4. It is evident from the unit cell diagram (Fig. 5) that this anionic layer is corrugated and the
unshared corner of each tetrahedron points towards the interlayer region which is occupied by $\mathrm{Cs}^{+}$ions. A similar corrugated anionic layer has been reported for $\mathrm{BaMoP}_{2} \mathrm{O}_{8}{ }^{19}$ with the yavapaiite structure. However, this type of layer is perfectly planar in phosphates like $\mathrm{ASbP}_{2} \mathrm{O}_{8}(\mathrm{~A}=\mathrm{K} \text { or } \mathrm{Tl})^{8,12}$ and other layered oxides such as $\mathrm{CsVMo}_{2} \mathrm{O}_{8}{ }^{20}, \mathrm{KAlMo}_{2} \mathrm{O}_{8}$ and $\alpha$ modifications of $\mathrm{AInMo}_{2} \mathrm{O}_{8}(\mathrm{~A}=\mathrm{Rb}$ or Cs$) .{ }^{17,21}$

Even though the $\mathrm{SbO}_{6}$ octahedron in compound 4 (Fig. 3) can be regarded as similar to those of $\mathbf{1}$ to $\mathbf{3}$, there is a wide variation of $\approx 0.065 \AA$ in $\mathrm{Sb}-\mathrm{O}$ bond lengths from 1.932 to 1.997 $\AA$ and a deviation of $\approx 5^{\circ}$ in $\mathrm{O}-\mathrm{Sb}-\mathrm{O}$ bond angles from the ideal values (Table 3). Thus the distortion of the $\mathrm{SbO}_{6}$ octahedron is more in 4 than in the other three. As in 1 to $\mathbf{3}, \mathrm{V}(1) \mathrm{O}_{4}$ and $\mathrm{V}(2) \mathrm{O}_{4}$ tetrahedra interconnect the chains and ribbons with the vanadium atoms, $\mathrm{V}(1)$ and $\mathrm{V}(2)$, forming short bonds with the unshared oxygen atoms, $\mathrm{O}(4)$ and $\mathrm{O}(8)$ respectively (Table 3). Even though the distribution of the values of the longer V-O bond lengths among these two types of tetrahedra in this compound is not as distinct as in $\mathbf{1}$ to $\mathbf{3}$, it is similarly more uniform in $\mathrm{V}(1) \mathrm{O}_{4}$ than in $\mathrm{V}(2) \mathrm{O}_{4}$. The vanadium atoms occupy the centres of both the tetrahedra.

We now refer to the oxygen co-ordinations of counter cations A, which are represented in Fig. 6. Potassium, rubidium and thallium present in the one dimensional channels of 1 to 3 are all eleven-co-ordinated and have the same type of co-ordination polyhedra. Cesium ions present in the interlayer


Fig. 5 Structure of $\mathrm{CsSbV}_{2} \mathrm{O}_{8}$. Isolated circles represent Cs atoms.
region of compound 4 can be considered as twelve-coordinated with one of the bonds namely, Cs-O(3), being very long ( $3.594 \AA$ ). It is to be noted that the co-ordination of Cs in other layered oxides, ${ }^{10,17,20,22}$ such as $\mathrm{CsVMo}_{2} \mathrm{O}_{8}$ containing a planar $\left(\mathrm{VMo}_{2} \mathrm{O}_{8}{ }^{-}\right)_{\infty}$ anionic layer, is more symmetrical. The average $\mathrm{A}-\mathrm{O}$ bond lengths for $\mathbf{1}$ to $\mathbf{4}$ are 3.020, 3.064, 3.062 and $3.358 \AA$ A respectively and these values are in accordance with the fact that the ionic radii of alkali metal ions increase from potassium to cesium and rubidium and thallium have almost the same ionic radii.

A comparison of the structural features of these four compounds and other $\mathrm{AMX}_{2} \mathrm{O}_{8}$ compounds $(\mathrm{A}=\mathrm{K}, \mathrm{Rb}, \mathrm{Cs}$ or Tl ; M with octahedral co-ordination; X with tetrahedral coordination) shows that Cs favours mostly layered structures whereas other monovalent A, smaller than Cs, adopt both layered and three dimensional structures. It is possible that the accommodation of larger Cs with higher co-ordination number is relatively easy in the layered structural frameworks. It is to be noted that the three dimensional structure of these vanadates has not been observed in the corresponding phosphates. ${ }^{8,12}$

## Vibrational spectroscopy

The infrared and Raman spectra of polycrystalline samples of compounds $\mathbf{1}$ to $\mathbf{4}$ show similar features (Fig. 7). A partial assignment of the modes is attempted here. The symmetric and antisymmetric stretching vibration modes of the $\mathrm{PO}_{4}{ }^{3-}$ group are generally observed between 1200 to 1300 and 1060 to 1160 $\mathrm{cm}^{-1}$ respectively. ${ }^{23}$ It has been established from a series of IR and Raman studies ${ }^{24}$ that the stretching frequencies of $\mathrm{SO}_{4}{ }^{2-}$ tetrahedra lie in the range 1200 to $950 \mathrm{~cm}^{-1}$. Accordingly, the range of vibrational stretching frequencies for $\mathrm{VO}_{4}{ }^{3-}$ of the present compounds can be tentatively assigned to the region 1022 to $920 \mathrm{~cm}^{-1}$. Similarly the bands from 300 to $200 \mathrm{~cm}^{-1}$ can be assigned to $\mathrm{V}-\mathrm{O}$ bending modes and those in the intermediate range of 500 to $400 \mathrm{~cm}^{-1}$ to the $\mathrm{Sb}-\mathrm{O}-\mathrm{V}$ stretching modes. Vanadates in solution, on the other hand, are known ${ }^{25}$ to show peaks around 827,780 and $340 \mathrm{~cm}^{-1}$ corresponding to $v_{1}, v_{3}$ and both $v_{3}$ and $v_{4}$ modes of tetrahedral $\mathrm{VO}_{4}{ }^{3-}$, respectively. Generally the $\mathrm{SbO}_{6}$ octahedra present typical bands at 900 to $800 \mathrm{~cm}^{-1}$ for antimony oxides. ${ }^{26}$ The $v_{1}, v_{2}$ and $v_{3}$ of $\mathrm{SbO}_{6}$ octahedra are reported to be present in the regions 830-680, $640-590$ and $570-480 \mathrm{~cm}^{-1}$ respectively. ${ }^{27}$ The symmetric and antisymmetric stretchings in the case of the $\mathrm{SbO}_{6}$ octahedra


Fig. 6 The ORTEP plots of oxygen co-ordinations of (a) potassium in 1, (b) thallium in 3 and (c) cesium in 4. Oxygen atoms are represented by circles of arbitrary radius and the metal atoms with $50 \%$ probability.
in the present compounds are, therefore, tentatively assigned to the region $920-600 \mathrm{~cm}^{-1}$. Bands in the range $400-300 \mathrm{~cm}^{-1}$ are attributed to the bending modes of $\mathrm{Sb}-\mathrm{O}$ and those in the range $200-80 \mathrm{~cm}^{-1}$ to the external modes.

## Conclusion

The four new vanadates of antimony, $\mathrm{ASbV}_{2} \mathrm{O}_{8}(\mathrm{~A}=\mathrm{K}, \mathrm{Rb}$, Tl or Cs ), prepared by solid state synthesis have been unambiguously characterized by single crystal X-ray diffraction. It is only the chemical vapour transport technique that has enabled the isolation of these materials in the form of single crystals suitable for X-ray diffraction. While the cesium compound is two dimensional in nature, the other three compounds have three dimensional frameworks with one dimensional channels. The counter cations $\mathrm{A}^{+}$occupy the channels and the interlayer region. The large size and requirement for higher co-ordination of cesium seems to favour the two dimensional structure. A partial assignment of the infrared and Raman spectra of the polycrystalline samples has been made. It would be interesting to synthesize and study similar vanadates of other pentavalent metals with octahedral co-ordination, such as niobium and tantalum. We have made some progress in this direction and the results will be published later.


Fig. 7 Infrared (left) and Raman (right) spectra of $\mathrm{ASbV}_{2} \mathrm{O}_{8}$ compounds, 1 to 4.

## Experimental

## Synthesis

The compounds $\mathbf{1}$ to $\mathbf{4}$ were prepared in the polycrystalline form by solid state reaction in open air; $\mathrm{KSbV}_{2} \mathrm{O}_{8}$ was prepared from a stoichiometric mixture of $\mathrm{KSb}(\mathrm{OH})_{6}$ and $\mathrm{V}_{2} \mathrm{O}_{5}$. The mixture was initially heated at $400^{\circ} \mathrm{C}$ for 12 h and then the temperature was raised in steps of $50^{\circ} \mathrm{C}$ to $600^{\circ} \mathrm{C}$ and the duration of heating at each step was about 9 h . Thus the reaction was carried out for a total duration of 2 d with three intermittent grindings. Similarly the other three compounds were prepared from stoichiometric mixtures of $\mathrm{Sb}_{2} \mathrm{O}_{5}, \mathrm{~V}_{2} \mathrm{O}_{5}$ and decomposable salts, $\mathrm{Rb}_{2} \mathrm{CO}_{3} / \mathrm{Cs}_{2} \mathrm{CO}_{3} / \mathrm{TlNO}_{3}$, at a maximum temperature of $700^{\circ} \mathrm{C}$.

## Crystal growth

Single crystals of the compounds were grown from their polycrystalline samples by chemical vapour transport (CVT) using $\mathrm{SbCl}_{5}$ as the transporting agent. The compound $\mathrm{SbCl}_{5}$ was initially sealed in thin glass capillary tubes of low melting $\left(\approx 300^{\circ} \mathrm{C}\right)$ glass. About 0.2 to 0.3 g of powdered $\mathrm{ASbV}_{2} \mathrm{O}_{8}$ compounds were sealed, along with sealed capillary tubes containing $\mathrm{SbCl}_{5}$, in an evacuated quartz tube of 1.3 cm diameter and 12 cm length. These quartz tubes were heated in a tubular furnace such that the sample was at the maximum temperature of its formation, $600 / 700^{\circ} \mathrm{C}$, and the other end of the tube was at a colder zone which was about $100{ }^{\circ} \mathrm{C}$ less. At $600 / 700^{\circ} \mathrm{C}$ the capillary tubes burst to provide $\mathrm{SbCl}_{5}$ vapour as transporting agent inside the sealed quartz tube. A bunch of transparent block shaped crystals grew at the colder end of the quartz tube after 4 d of heating.

## X-Ray diffraction and crystal structure

The powder X-ray diffraction (XRD) patterns of polycrystalline samples of compounds $\mathbf{1}$ to $\mathbf{4}$ were recorded on Rigaku Miniflex diffractometer using Ni-filtered Co-K $\alpha$ (1.7902 Å) radiation. Silicon was used as external standard.

Single crystals suitable for X-ray diffraction were isolated from the bunch of the crystals grown by the CVT method.

Diffraction data for the single crystals of the compounds $\mathbf{1}$ to 4 were collected at $293 \pm 2 \mathrm{~K}$ on an Enraf-Nonius CAD4 diffractometer. There was no detectable decay during the data collection according to the intensities of three reflections measured at regular intervals. Absorption corrections based on azimuthal scans of reflections with angle $\chi \approx 90^{\circ}$ were applied to all the four data sets. The space groups were determined from systematic absences. For compounds $\mathbf{1}$ and 2 the centrosymmetric Pnma was chosen over non-centrosymmetric Pna2 ${ }_{1}$. The choice was found to be correct by the successful refinement. The structure solution and refinement were done using the programs SHELXS 86 and SHELXL 93 respectively. ${ }^{28}$ The metal atoms were first located by direct methods. Refinement of their positions and subsequent Fourier difference maps led to the location of oxygen atoms. All the atoms were refined by full matrix least squares based on $F^{2}$. For compounds $\mathbf{1}$ and $\mathbf{2}$ the anisotropic refinement of all the atoms of the asymmetric unit proceeded smoothly to give acceptable $R$ values and the final Fourier difference maps did not have chemically significant features. For 3 all the atoms except one vanadium atom, $V(1)$, were refined anisotropically. For 3 and 4 the $R$ values were found to be 0.044 and 0.054 respectively and the final Fourier difference maps showed about 20 peaks with electron density $>1$ e $\AA^{-3}$. These peaks were found to be ghosts of existing atoms indicating severe absorption problems. Therefore an additional absorption correction using the DIFABS program ${ }^{29}$ was applied to isotropically refined data sets for these two compounds. The final anisotropic refinement carried out using the corrected data of the two compounds led to improved $R$ values (Table 1) and reduced electron density of ghost peaks. Several attempts were made to grow better crystals of $\mathbf{3}$ and 4 but were unsuccessful.

CCDC reference number 186/1081.
See http://www.rsc.org/suppdata/dt/1998/3013/ for crystallographic files in .cif format.

## Spectroscopy

The infrared spectra of these compounds were recorded on a Bruker 17S 66V FT-IR spectrometer. The samples were ground with dry KBr and pressed into transparent discs. The Raman spectra were recorded on an RFS 100 FT Raman spectrometer with a FRA 106 FT Raman accessory provided with a $\mathrm{CaF}_{2}$ beam splitter; a Nd:YAG laser operating at an output of 200 mW for a 1064 nm line was used as the excitation source.

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## References

1 B. Raveau, M. M. Borel, A. Leclaire and A. Grandin, Int. J. Mod. Phys. B, 1993, 7, 4109
2 G. Constentin, A. Leclaire, M. M. Borel, A. Grandin and B. Raveau, Rev. Inorg. Chem., 1993, 13, 77.

3 M. M. Borel, M. Goreaud, A. Grandin, Ph. Labbe, A. Leclaire and B. Raveau, Eur. J. Solid State Inorg. Chem., 1991, 28, 93.
4 B. Raveau, Proc. Indian Acad. Sci. (Chem. Sci.), 1986, 96, 419.
5 M. Goreaud, Ph. Labbe and B. Raveau, J. Solid State Chem., 1985, 56, 41.
6 Y. Piffard, A. Lachgar and M. Tournoux, J. Solid State Chem., 1985, 58, 253.
7 L. O. Hagman and P. Kierkegaard, Acta Chem. Scand., 1968, 22, 1822.

8 Y. Piffard, S. Oyetola, S. Courant and A. Lachgar, J. Solid State Chem., 1985, 60, 209.
9 M. Fakhfakh, S. Oyetola, N. Jouini, A. Verbaere and Y. Piffard, Mater. Res. Bull., 1994, 29, 97.

10 S. Oyetola, A. Verbaere, Y. Piffard and M. Tournoux, Eur. J. Solid State Inorg. Chem., 1988, 25, 259.
11 V. P. Nikolaev, G. G. Sadikov, A. V. Lavrov and M. A. PoraiKoshits, Dokl. Akad. Nauk SSSR, 1982, 264, 859.
12 F. Zid, T. Jouini and N. Jouini, C. R. Acad. Sci., Ser. 2, 1989, 309, 29.

13 S. A. Linde, Yu. E. Gorbunova, A. V. Lavrov and I. V. Tananaev, Sov. Phys. Dokl., 1980, 25, 4.
14 J. M. Troup and A. Clearfield, Inorg. Chem., 1977, 16, 3311.
15 V. B. Zlokazov, J. Appl. Crystallogr., 1992, 25, 69.
16 R. F. Klevstova and P. V. Klevstov, Sov. Phys. Crystallogr., 1971, 16, 241.
17 V. A. Efremov, V. K. Trunov and Yu. A. Velikodnyi, Russ. J. Inorg. Chem., 1971, 16, 560.
18 C. K. Johnson, ORTEP, Report ORNL-5138, Oak Ridge National Laboratory, Oak Ridge, TN, 1976.
19 A. Leclaire, M. M. Borel, J. Chardon and B. Raveau, J. Solid State Chem., 1995, 116, 364.
20 K. H. Lii, C. C. Wang, R. K. Chiang and S. L. Wang, J. Solid State Chem., 1989, 80, 144.

21 V. A. Efremov, V. K. Trunov and Yu. A. Velikodnyi, Sov. Phys. Crystallogr:, 1973, 17, 1005.
22 R. W. Berg, S. Boghosian, N. J. Bjerrum, R. Fehrmann, B. Krebs, N. Strater, O. S. Mortensen and G. N. Papatheodorou, Inorg. Chem., 1993, 32, 4714 and refs. therein.
23 I. L. Botto and A.C. Garcia, Mater. Res. Bull., 1989, 24, 1431.
24 R. Fehrmann, B. Krebs, G. N. Papatheodorou, R. W. Berg and N. J. Bjerrum, Inorg. Chem., 1986, 25, 1571.

25 W. P. Griffith and T. D. Wickins, J. Chem. Soc. A, 1966, 1087.
26 M. Jansen, J. Pebler and K. Dehnicke, Z. Anorg. Allg. Chem., 1982, 495, 120.
27 E. Husson, Y. Repelin and H. Brusset, Spectrochim. Acta, Part A, 1979, 35, 1177.
28 G. M. Sheldrick, SHELXS 86, University of Göttingen, 1986; SHELXL 93, University of Göttingen, 1993.
29 N. Walker and D. Stuart, Acta Crystallogr., Sect. A, 1983, 39, 159.

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[^0]:    $\dagger$ Supplementary data available: XRD powder data. For direct electronic access see http://www.rsc.org/suppdata/dt/1998/3013/, otherwise available from BLDSC (No. SUP 57412, 8 pp.) or the RSC Library. See Instructions for Authors, 1998, Issue 1 (http://www.rsc.org/dalton).

